# Identifying Electronic Properties Relevant to Improving Stability in a-Si:H-Based Cells and Overall Performance in a-Si,Ge:H-Based Cells

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#### **PREFACE**

This Final Technical Progress Report covers the work performed at the University of Oregon for the period 18 April 1994 to 15 January 1998 under NREL Subcontract Number XAN-4-13318-07. The following personnel participated in this research program:

NAME	TITLE	WORK PERFORMED
J. David Cohen	Principal Investigator	Program Manager
Chintimani Palsule	Research Associate	Characterization of the hetero- junction interface between a-Si:H and a-Si,Ge:H Alloys
Yoram Lubianiker	Research Associate	Stability of hydrogen diluted and ECR deposited a-Si:H
Fan Zhong	Research Assistant	Defect properties and stability studies of a-Si,Ge:H Alloys
Chih-Chiang Chen	Research Assistant	Defect properties and stability studies of a-Si,Ge:H Alloys; Studies of hydrogen diluted a-Si:H films.
Daewon Kwon	Research Assistant	Studies of Hot-Wire deposited a-Si:H films

#### **TABLE OF CONTENTS**

		Page
LIST	T OF ILLUSTRATIONS	iv
LIST	T OF TABLES	vii
EXE	CUTIVE SUMMARY	viii
1.0	INTRODUCTION	1
2.0	SAMPLES AND SAMPLE TREATMENT	
3.0	2.1 GLOW DISCHARGE AMORPHOUS SILICON- GERMANIUM ALLOYS	2 4 5 5
	3.1 ADMITTANCE SPECTROSCOPY	8
4.0	ELECTRONIC PROPERTIES OF THE UNI-SOLAR a-Si,Ge:h ALLOYS	14
	4.1 DEEP DEFECT PROPERTIES AND DEGRADATION	
5.0	CATHODIC AMORPHOUS SILICON-GERMANIUM ALLOYS	24
	5.1 MOBILITY GAP STRUCTURE AND DEEP DEFECTS	
6.0	PROPERTIES OF THE a-Si:H/a-Si,Ge:H HETEROJUNCTION INTERFACE	34

7.0	HOT-WIRE DEPOSITED AMORPHOUS SILICON	39
8.0	STABILITY OF HYDROGEN DILUTED GLOW DISCHARGE a-Si:H	43
	8.1 PROPERTIES OF LBL a-Si:H DEPOSITED WITH HYDROGEN OR HELIUM DILUTION	43
	8.2 PROPERTIES OF SOLAREX a-Si:H DEPOSITED WITH HYDROGEN DILUTION	
	8.3 POSSIBLE MECHANISMS: HYDROGEN DILUTION MODULATION STUDIES	47
9.0	STUDIES OF ELECTRON-CYCLOTRON RESONANCE DEPOSITED AMORPHOUS SILICON	50
10.0	SUMMARY AND CONCLUSIONS	
11.0	SUBCONTRACT SUPPORTED PUBLICATIONS	54
12.0	REFERENCES	55

#### LIST OF ILLUSTRATIONS

		Page
FIG. 1.	Drive-level capacitance profiles for an a-Si,Ge:H sample	9
FIG. 2	Summary of drive-level densities vs. emission energy for three a-Si,Ge:H samples	10
FIG. 3.	Schematic diagram indicating the basic sequence of events in semiconducting junction transient measurements	11
FIG. 4.	Schematic diagram of the types of optically initiated defect transitions	13
FIG. 5.	Bar graph indicating the magnitudes of the DLCP determined TH1 band for 8 a-Si,Ge:H samples	14
FIG. 6.	Pairs of photocapacitance and photocurrent spectra for three a-Si,Ge:H samples containing 35at.% Ge	15
FIG. 7.	Energy distributions of defect bands used to fit the three sample spectra of Fig. 6.	17
FIG. 8.	Transient photocapacitance and photocurrent spectra for the annealed and light soaked state of an intrinsic 35at.% sample	17
FIG. 9.	Densities of states employed to fit sub-band-gap optical spectra of Fig. 8	18
FIG. 10.	Ratios of the magnitudes of the two defect bands, OP2/OP1 derived form analysis of the optical spectra	19
FIG. 11.	Ratios of the magnitudes of the OP2 (D <sup>+</sup> ) band to TH1	19
FIG. 12.	<ul> <li>(a) Dependence of the hole μτ products deduced from the analysis of the photocapacitance/photocurrent ratios on alloy composition</li> <li>(b) Variation of (μτ)<sub>h</sub> TH1 with alloy composition</li> </ul>	22
FIG. 13.	Variation of the ambipolar diffusion lengths and the corresponding defect densities for two Uni-Solar samples in states A and B	23
FIG. 14.	Product of the ambipolar diffusion length and the square root of the deep defect density	23
FIG. 15.	Temperature dependence of the drive-level density of a cathode deposited sample and of a Uni-Solar sample	25

FIG. 16.	Comparison of the total defect densites vs. Ge content for the a-Si,Ge:H alloys from three sources	25
FIG. 17.	Transient photocurrent sub-band-gap spectra for 3 Harvard cathodic samples together with the spectra of three of the Uni-Solar a-Si,Ge:H samples	26
FIG. 18.	Pairs of photocapacitance and photocurrent transient spectra for two a-Si,Ge:H samples taken at 330K	27
FIG. 19.	Energy positions of the defect bands deduced from the sub-band-gap optical spectra for three series of a-Si,Ge:H samples	28
FIG. 20.	Measured vs. calculated values of deep defect densities for a-Si,Ge:H samples from various sources	30
FIG. 21.	<ul> <li>(a) Variation of the ambipolar diffusion length and defect density vs. the air contamination level in four a-Si,Ge:Hsamples</li> <li>(b) Product of the ambipolar diffusion length and the square root of the defect density</li></ul>	33
FIG. 22.	Capacitance vs. time following a 300s long voltage filling pulse for one a-Si:H/a-Si,Ge:H heterojunction sample	34
FIG. 23.	Capacitance overshoot vs. the product of the filling pulse voltage and the filling pulse duration for heterojunction sample 6	35
FIG. 24.	Capacitance transents for heterojunction sample 7 using different ambient bias voltages	37
FIG. 25.	Charge transients deduced from capacitance transient measurements on a-Si:H/a-Si,Ge:H heterojunction samples	37
FIG. 26.	Comparison of the transient photocapacitance for hot-wire a-Si:H samples for 3 different substrates temperatures	39
FIG. 27.	Variation of the optical gaps and the Urbach energy vs. hydrogen content for the hot-wire a-Si:H samples	40
FIG. 28.	Drive-level capacitance profiling data for three hot-wire a-Si:H samples before and after light soaking	40
FIG. 29.	Comparison of the total deep defect obtained both from the capacitance profiling data and the sub-band-gap photocapacitance spectra	41
FIG. 30.	Comparison of defect densities before and after light soaking	42

FIG. 31.	Photocapacitance and transient photocurrent spectra for sample HW360 in its annealed and light-soaked states	42
FIG. 32.	Example of drive-level capacitance profiling data for hydrogen diluted a-Si:H sample	44
FIG. 33.	Comparison of drive-level densities for different contacting configurations for two types of a-Si:H films	44
FIG. 34.	Comparison of drive-level determined defect densities for one hydrogen diluted and an undiluted sample vs. light exposure	45
FIG. 35.	Drive-level deep defect profiles for one Solarex hydrogen diluted a-Si:H sample and one undiluted sample in their light degraded states	46
FIG. 36.	Photocapacitance sub-band-gap spectra for the same two samples as in Fig. 35 in their light degraded states	47
FIG. 37.	Drive-level capacitance profiling data for three light soaked states and three post annealed states of H <sub>2</sub> diluted, Ar diluted and modulated a-Si:H films	49
FIG. 38.	Photocapacitance sub-band-gap spectra for a H2 diluted ECR deposited a-Si:H in its as-grown state and after 50h of intense light degradation	51

### LIST OF TABLES

TABLE I.	Optical and electrical properties of the first series of cathode deposited glow discharge a-Si <sub>1-x</sub> Ge <sub>x</sub> :H alloy samples	3
TABLE II.	Impurity levels of the second series, air contaminated cathodic a-Si,Ge:H samples	3
TABLE III.	Summary of configurations for a-Si:H/a-Si,Ge:H heterojunction samples	4
TABLE IV.	Characterisitics of hot-wire deposited a-Si:H samples	5
TABLE V.	Deposition conditions for the i-layers of a-Si:H samples produced at Lawrence Berkeley Laboratory	6
TABLE VI.	Defect sub-band densites of Uni-Solar a-Si,Ge:H alloys determined by C-T-ω, DLCP, transient photocapacitance and photocurrent methods	19
TABLE VII.	Properties of two Uni-Solar a-Si,Ge:H alloy samples in both their annealed and degraded conditions	23
TABLE VIII.	Basic transport properties of the air contaminated cathodic a-Si,Ge:H samples in state A	31
TABLE IX.	Defect densities determined from DLCP and from fits to the subband-gap spectra for the cathodic a-Si,Ge:H samples	32
TABLE X.	Summary of deduced interface trap densities for a-Si:H/a-Si,Ge:H samples	38
TABLE XI.	Carrier mobilities and deep defect densities for LBL samples in their dark annealed states	45
TABLE XII	Summary of preliminary reults for ECR a-Si:H properties	51

#### **EXECUTIVE SUMMARY**

The work carried out under NREL Subcontract XAN-4-13318-07 has been focused on the characterization and evaluation of low gap (a-Si,Ge:H) alloy materials and on issues related to overall stability in the mid-gap (a-Si:H) materials. First of all, we characterized a extensive series of Uni-Solar a-Si,Ge:H samples using drive-level capacitance profiling (DLCP) and the analysis of sub-band-gap photocapacitance and photocurrent spectra. We thus identified several bands of deep defect transitions. We had hypothesized that one type of defect band optical transition in these alloys, that did not seem to be present in pure a-Si:H, might indicate a significant population of D<sup>+</sup> states in these nominally intrinsic a-Si,Ge:H samples. We also carried out light induced degradation studies of these alloys which tended to support this conclusion and indicated that charged defect ratios could vary significantly after light soaking. We then examined the properties of very lightly n- and p-type a-Si,Ge:H material which verified that charged defects are indeed responsible for the different observed defect bands in device quality a-Si,Ge:H alloy material. This conclusion undoubtedly will have important consequences for understanding the transport and degradation process in a-Si,Ge:H devices.

Second, we reported results of our measurements on a-Si,Ge:H alloy "cathodic" samples produced at Harvard University. These samples were found to exhibit significantly lower defect densities in the high Ge composition range (>50at.% Ge) than alloy samples produced either by conventional glow discharge or photo-CVD deposition. Moreover, this lower defect density appeared to be entirely consistent with simple defect formation models given the differences observed for other aspects of the electronic structure in these samples: a larger gap energy for a given Ge fraction, a different relative energy position of the defect within the gap, and a smaller Urbach energy. However, our measurements also indicated a much smaller value of  $(\mu\tau)_h$  for these samples than would have been expected given their lower defect densities.

Third, we performed voltage pulse stimulated capacitance transient measurements on a-Si:H/a-Si,Ge:H heterostructure samples to look for carrier trapping states that might be associated with this interface. We found there was a clear signature of trapped hole emission extending over long times associated specifically with the interface itself in concentrations of roughly  $10^{11} \, \mathrm{cm}^{-2}$ . However, we found that these hole traps did not seem to act as recombination centers for electrons brought into the interface region. Nonetheless, these traps seem to exist in sufficient densities to alter the electric field profiles across such heterojunction structures and, therefore, impact the performance of cells which incorporate such interfaces.

Fourth, we reported our results on several hot-wire a-Si:H samples produced with varying hydrogen levels. These samples were evaluated in both their as-grown state as well as the light degraded state. We found that samples with a H content above 10at.% exhibited essentially identical properties to those of conventional glow discharge a-Si:H. However, as the H level was decreased to about 2at.% the electronic properties actually *improved*: the degraded defect level was reduced and the Urbach tail was slightly narrower. These changes were accompanied by more than a 0.1eV decrease in optical gap. Therefore, our studies indicate that hot-wire produced a-Si:H, with H levels between 2-5at.%, should lead to mid-gap devices with superior properties.

Finally, we discussed some results on glow discharge material as well as ECR deposited a-Si:H grown under hydrogen dilution conditions. We confirmed that, in terms of deep defect creation, such films exhibited improved stability compared to conventional glow discharge material: roughly a factor of three lower deep defect densities than those grown using pure silane. Furthermore, the hydrogen diluted samples degraded at a slower rate and saturated at a significantly lower value (by about a factor of five) than pure silane deposited sample. These results agree with reports of increased relative stability of cells employing hydrogen-diluted i-layers. To try to gain some insight into the mechanisms responsible for such differences in stability we also compared Ar and H diluted SiH<sub>4</sub> grown samples with a sample that was switched periodically between these two types of gas mixtures during growth. While still very preliminary, our studies indicate that film strain may play an important role.